



UNIVERSITÀ
DEGLI STUDI
FIRENZE
DIEF
DIPARTIMENTO
DI INGEGNERIA
INDUSTRIALE

October 17th–18th, 2022

MICROSTRUCTURAL AND MECHANICAL CHARACTERIZATION OF NITINOL STENTS PRODUCED BY LASER POWDER BED FUSION BEFORE AND AFTER HEAT TREATMENT

Maria Beatrice Abrami

Università degli Studi di Brescia, via Branze 38, Brescia ING-IND/21

📍 Plesso Didattico Morgagni, Viale Morgagni,
44-48, 50134 Firenze

ADDITIVE 4 BIOMEDICAL



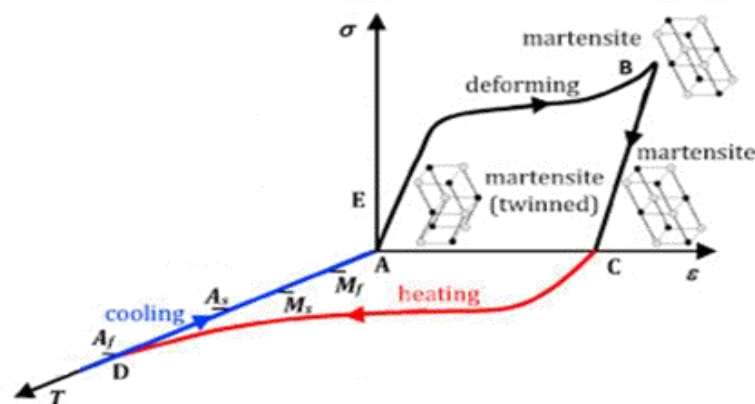
UNIVERSITÀ
DEGLI STUDI
DI BRESCIA



NITINOL (Ni-Ti)

- Near-equiatomic nichel-titanium alloy, the most common shape memory metal material
- Two different phases depending on T which determine two different behaviors:

SHAPE MEMORY EFFECT



Martensite

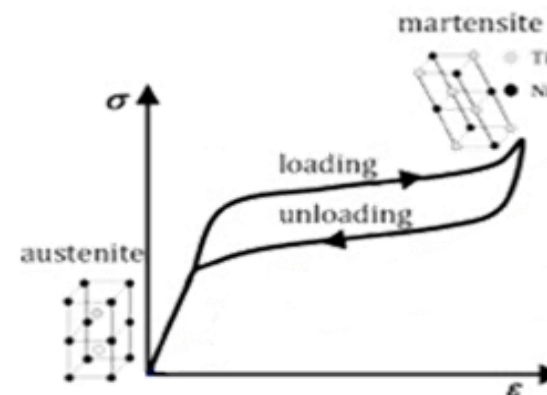
$$M_f > T_{\text{operation}}$$

Transformation
Temperatures (TTs)

> %Ti

Temperature sensors,
actuators...

SUPERELASTICITY



Austenite

$$A_f < T_{\text{operation}}$$



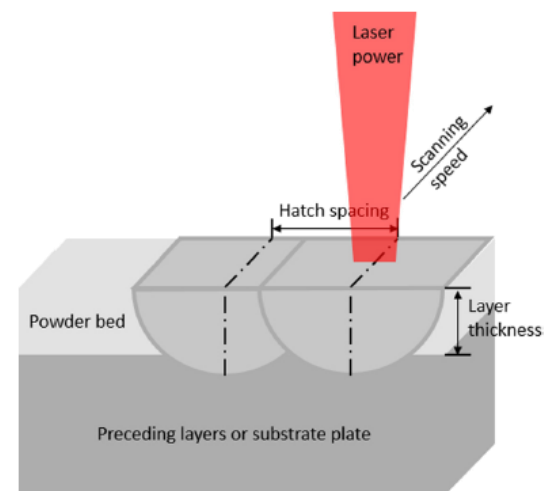
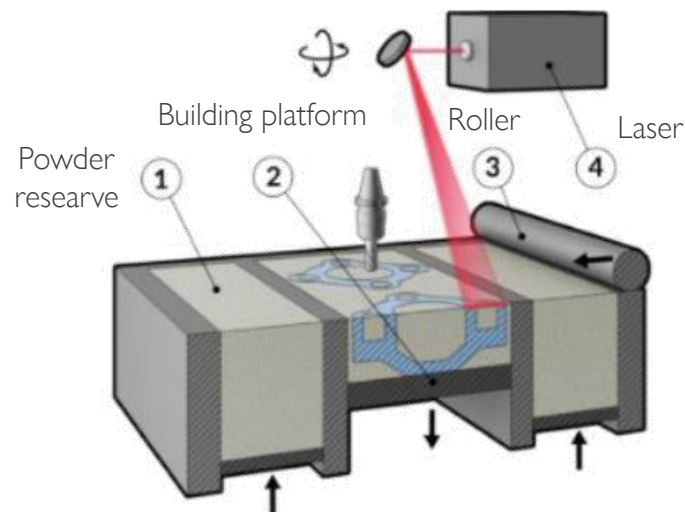
> %Ni

Biomedical implants: stents,
filters, orthopedic implants...



LASER POWDER BED FUSION (L-PBF)

- Additive manufacturing technique which consists in manufacturing components by adding material layer by layer



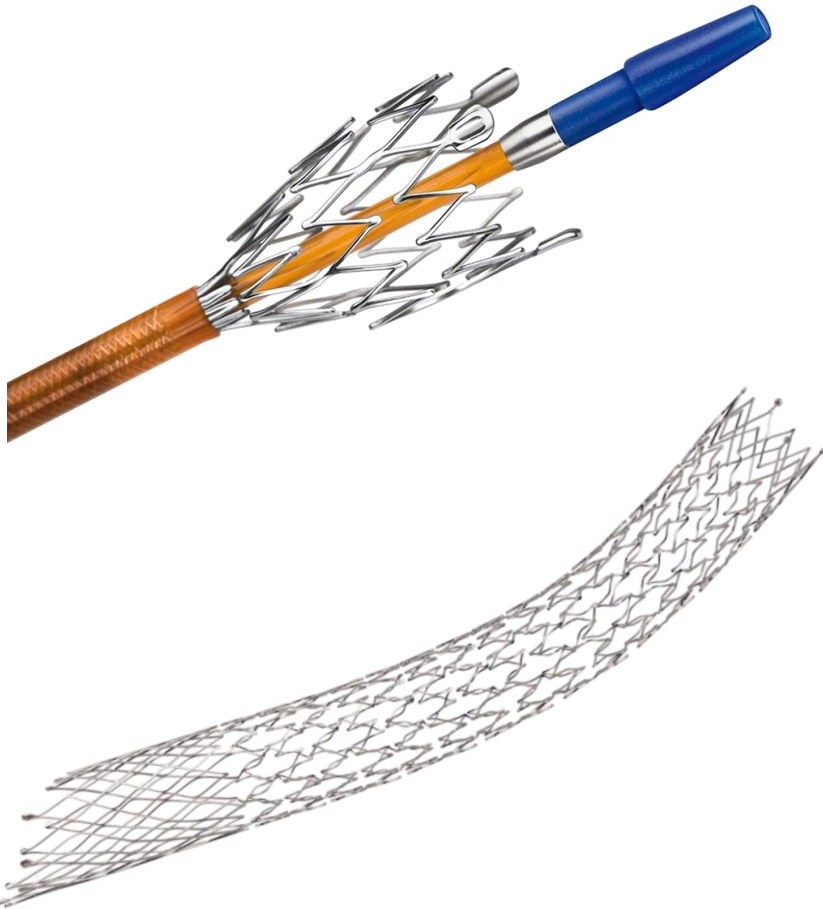
- In the **biomedical field**:

✓ Advantages	✗ Limits
Freedom of design	Process complexity
Possibility to manufacture tailor-made biomedical implants with complex shape	Wide number of parameters difficult to handle
Lower costs	Typical process defects

- For **Nitinol**:

- Ni evaporation during laser processing
- Formation of secondary phases
- Temperature transformation control
- **Obtainment of the superelasticity**

AIM OF THE WORK



Investigate the possibility to use **additive manufacturing** techniques to produce **self-expandable stents** in Nitinol

Process feasibility
Parameters
Nichel evaporation

Resulting properties
Transformation Temperatures
Microstructure
Mechanical properties

Overcome the limits of the standardized conventional production

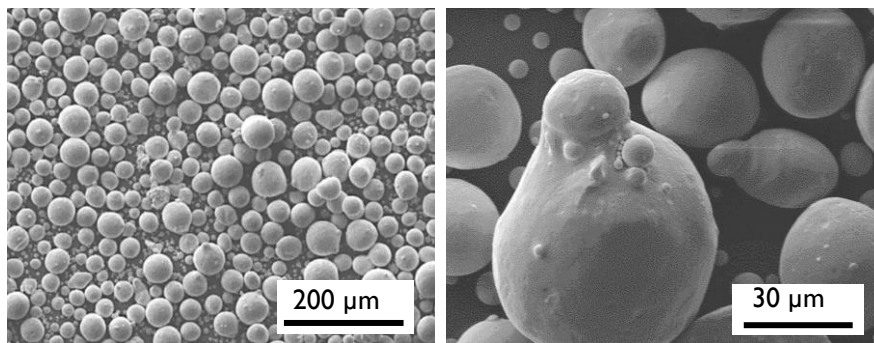
EXPERIMENTAL PROCEDURE

➤ Production



Nitinol powder (%at)

Ni	Ti
50.54	49.46

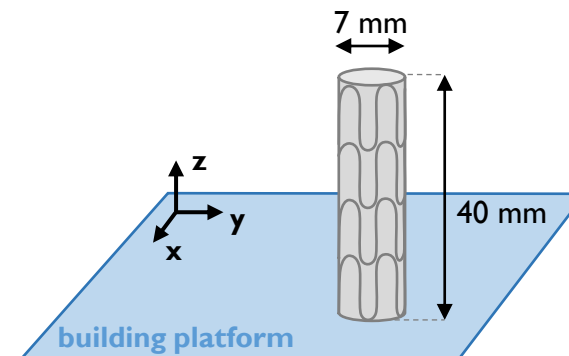


10-70 μm

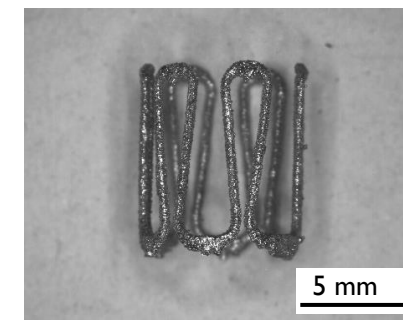
Laser powder bed fusion			
Sample	Laser Power	Scan Speed	Hatch Spacing
A	150 W	900 mm/s	55 μm
B	180 W	1200 mm/s	70 μm
C	180 W	900 mm/s	70 μm

Layer thickness = 40 μm

Spot size = 40 μm



Cutting



➤ Heat treatments

T1	600 °C	1.5 h
T2	1100 °C	1.5 h

- High vacuum
- Argon partial pressure 1 mbar

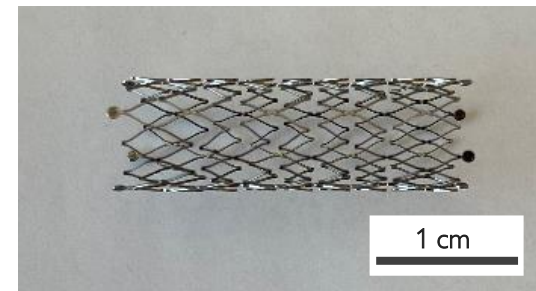


VACUUM FURNACES

EXPERIMENTAL PROCEDURE

➤ Conventional stent

- NiTi, self-expandable
- Superficial Femoral Artery Stent (SFA Stent)
- Medtronic



➤ Characterization: as-built (A, B, C), T1, T2, SFA stent

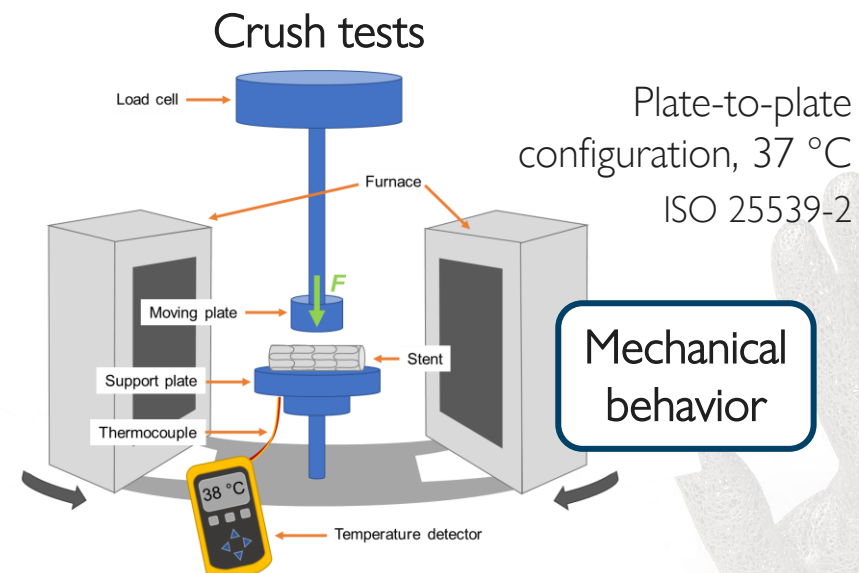
Optical microscope
SEM

Microstructural
characterization

DSC analyses

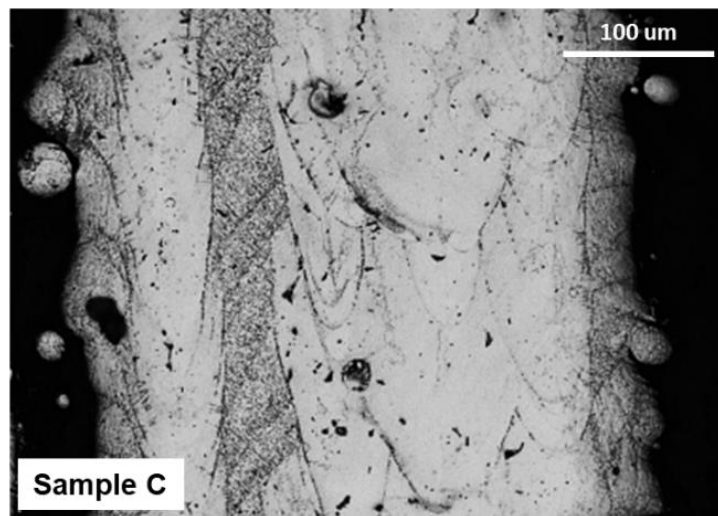
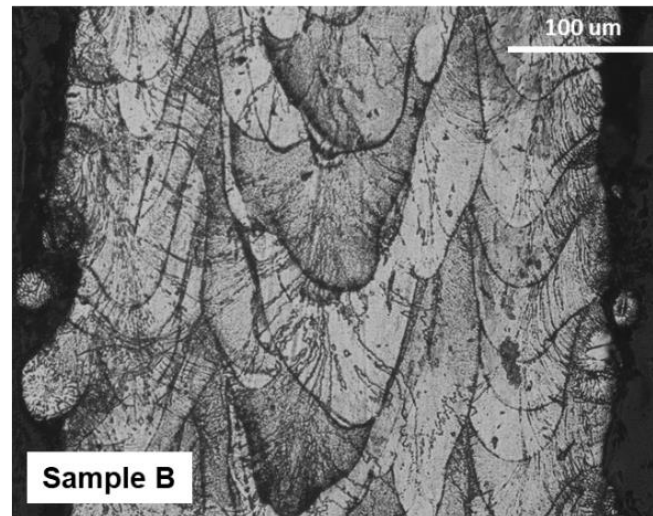
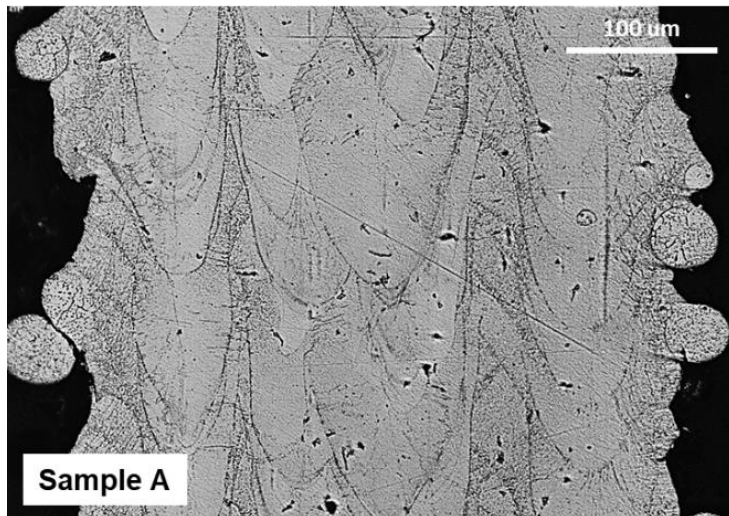
-90 ÷ 180 °C
10 °C/min

Transformation
Temperatures (TTs)



RESULTS: AS-BUILT SAMPLES

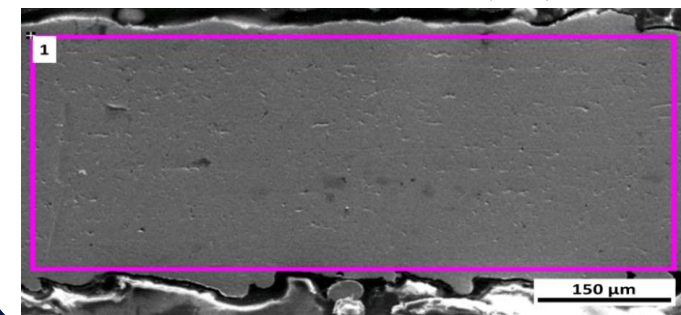
- Melt pools and columnar grains
- Porosities
- Not completely fused particles → poor surface finishing



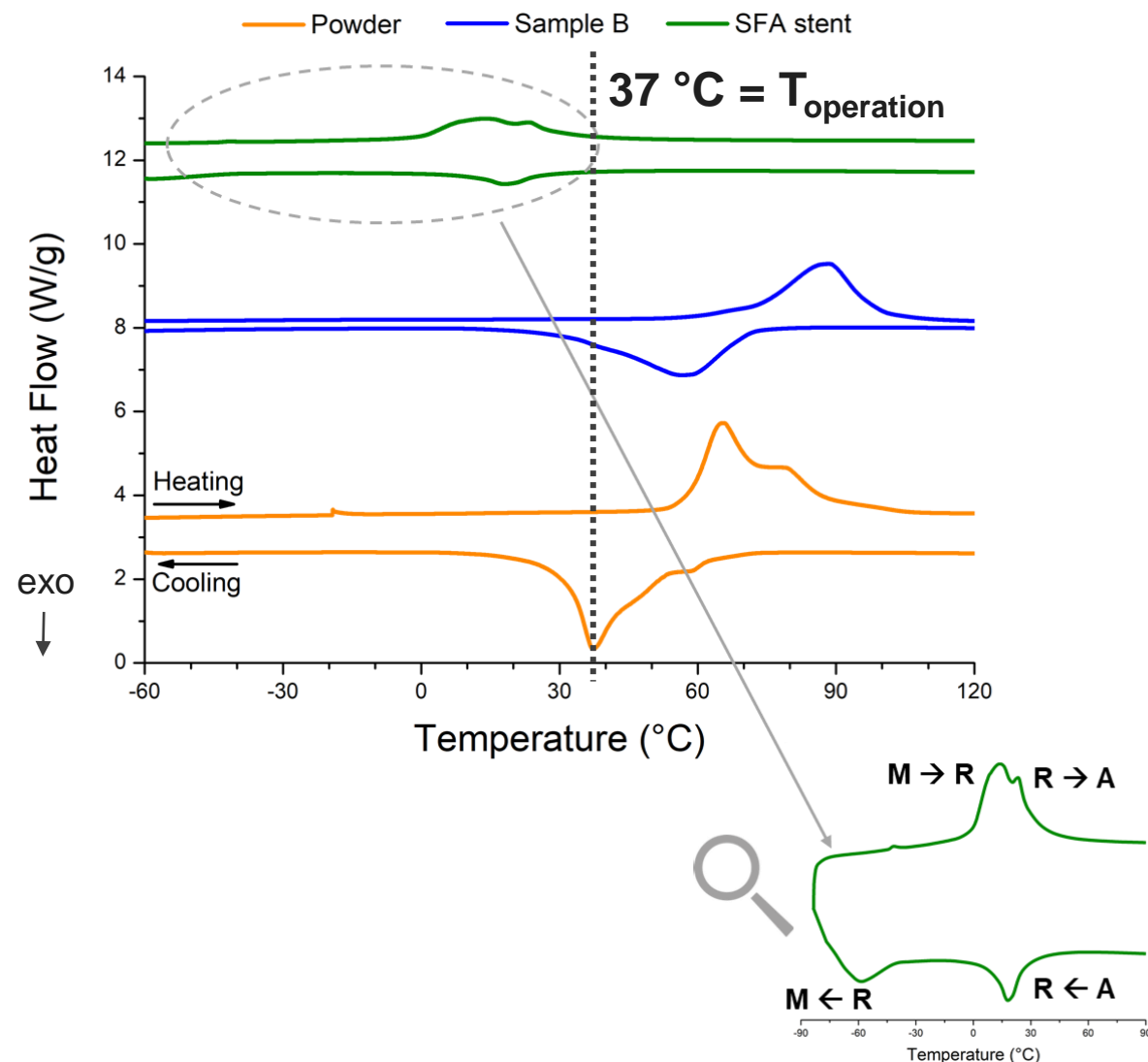
1.5% Ni loss

	Ti	Ni
1	51.02	48.98

(at%)



RESULTS: AS-BUILT SAMPLES



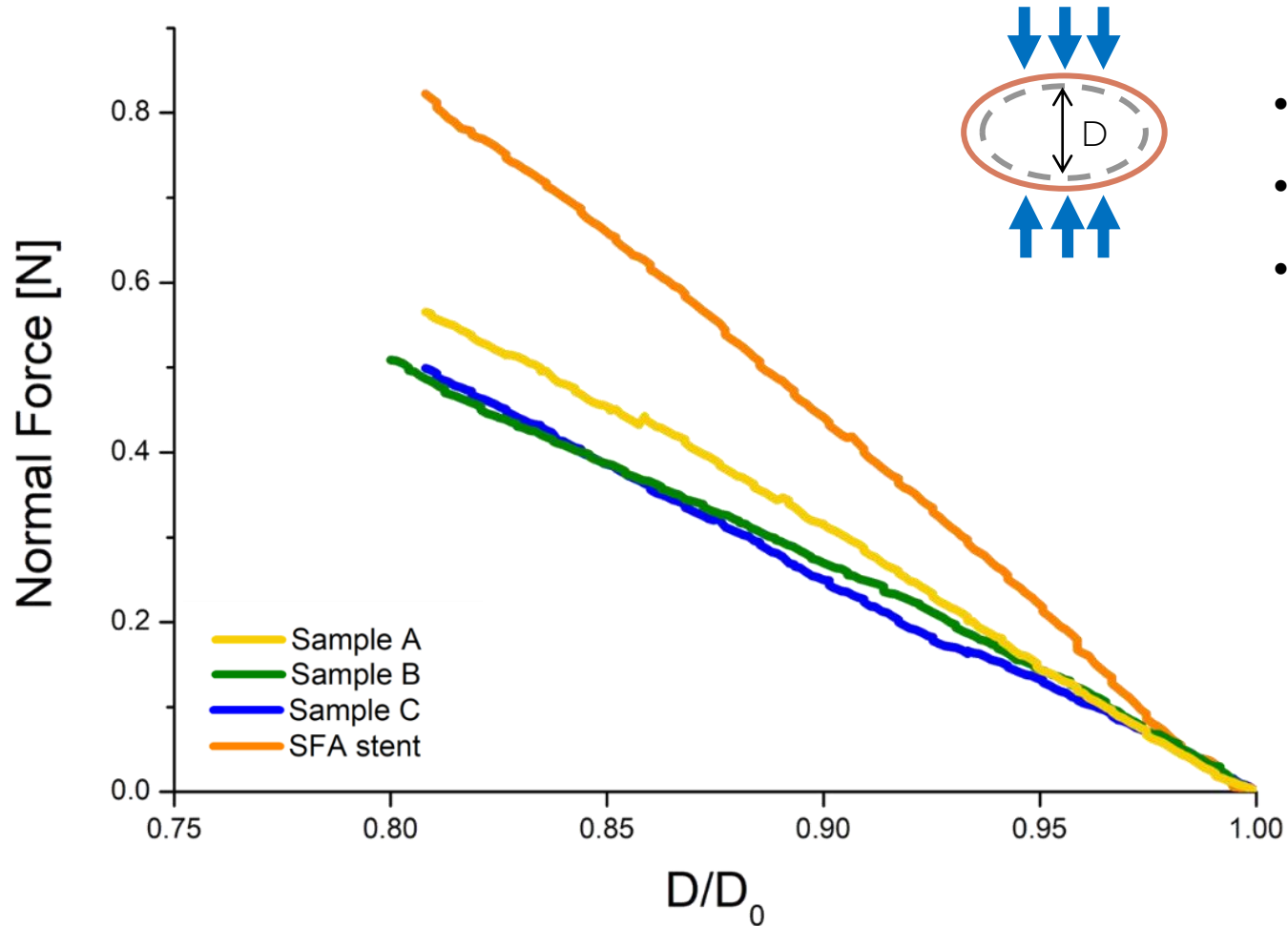
SFA stent

- Double peaks
 - $|T_A - T_R| < 10 \text{ °C}$
 - Austenite at $T_{\text{operation}}$
- presence of the R-phase

L-PBF samples

- Equal TTs in samples A, B, C
- No double peaks
- Martensite at $T_{\text{operation}}$
- Curves shift to the right compared to powder → Ni loss during the process

RESULTS: AS-BUILT SAMPLES



- Almost linear behaviour
- Negligible effect of the process parameters
- Lower compressive strength of L-PBF stents compared to SFA stents

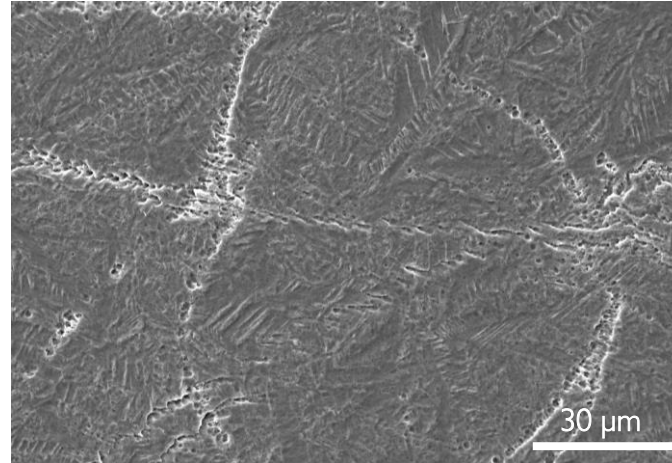
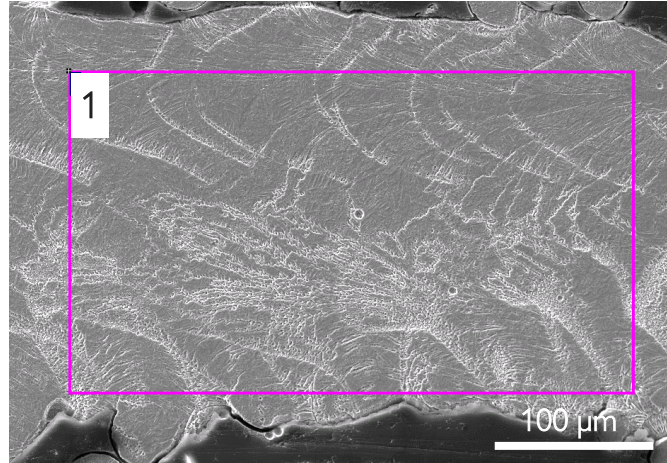
Compression resistance (N)

Sample A	0.70 ± 0.19
Sample B	0.51 ± 0.07
Sample C	0.45 ± 0.07
SFA stent	0.82 ± 0.03



RESULTS: HEAT-TREATED SAMPLES

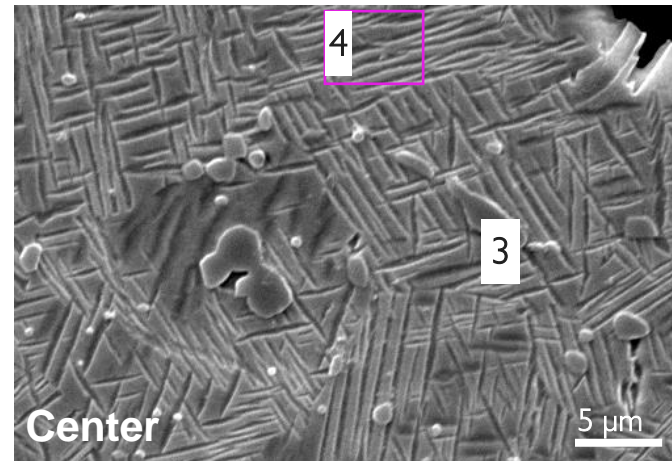
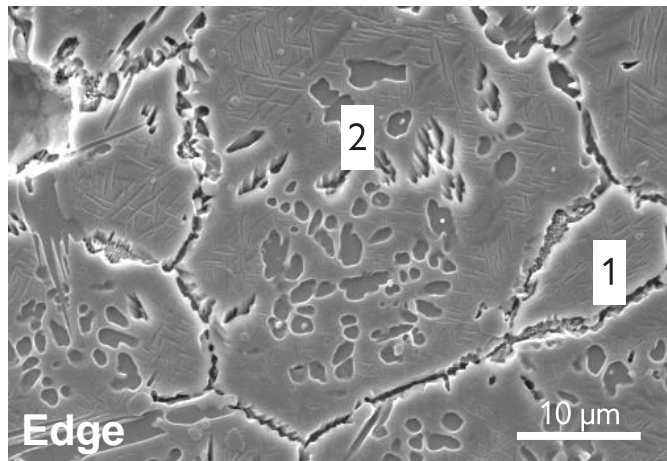
T1



	Ti	Ni
1	50.99	49.07

(at%)

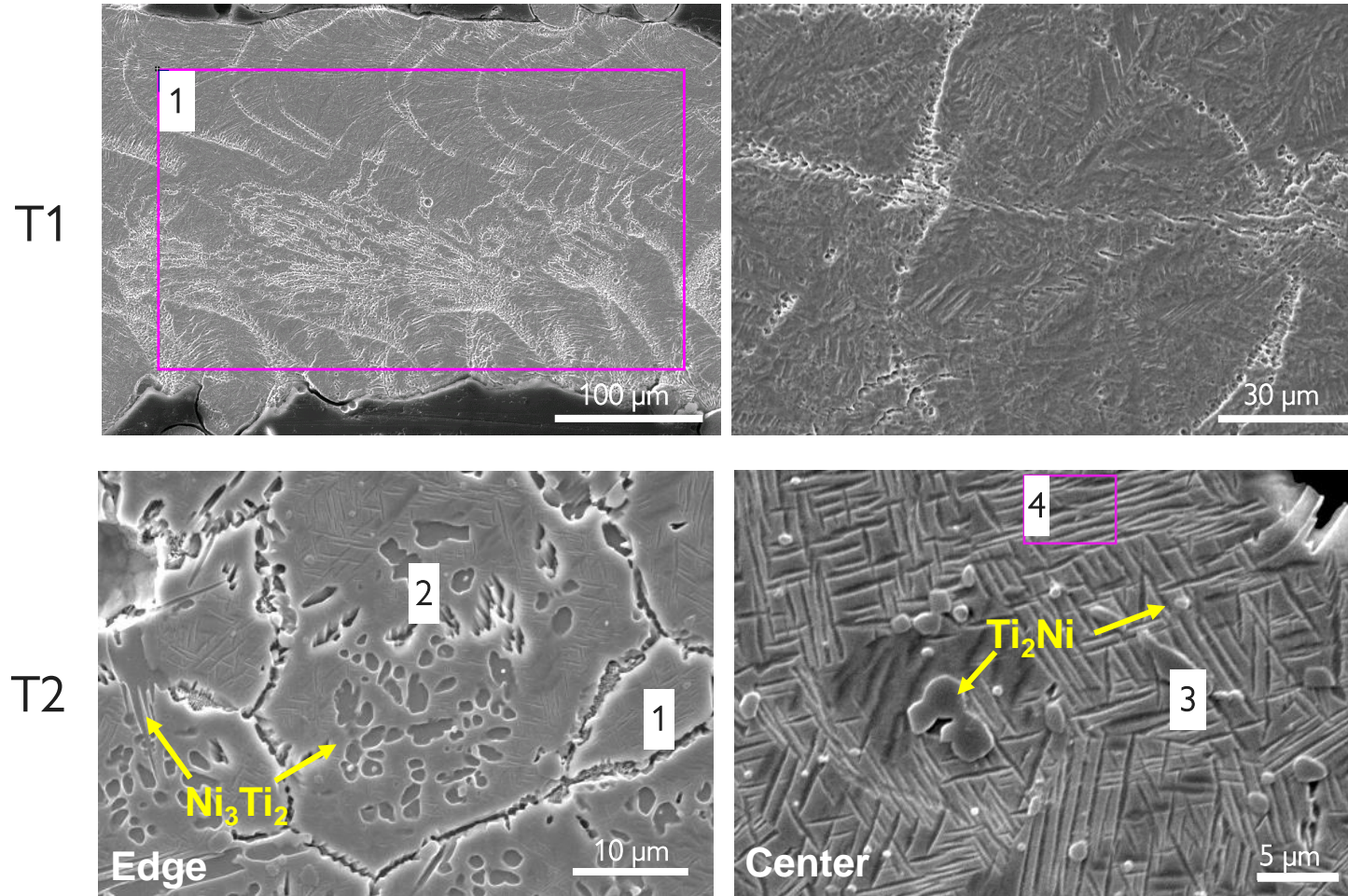
T2



	Ti	Ni	
1	49.49	50.51	→ matrix
2	41.86	58.14	→ Ni-rich
3	63.84	36.16	→ Ti-rich
4	48.23	51.77	→ matrix

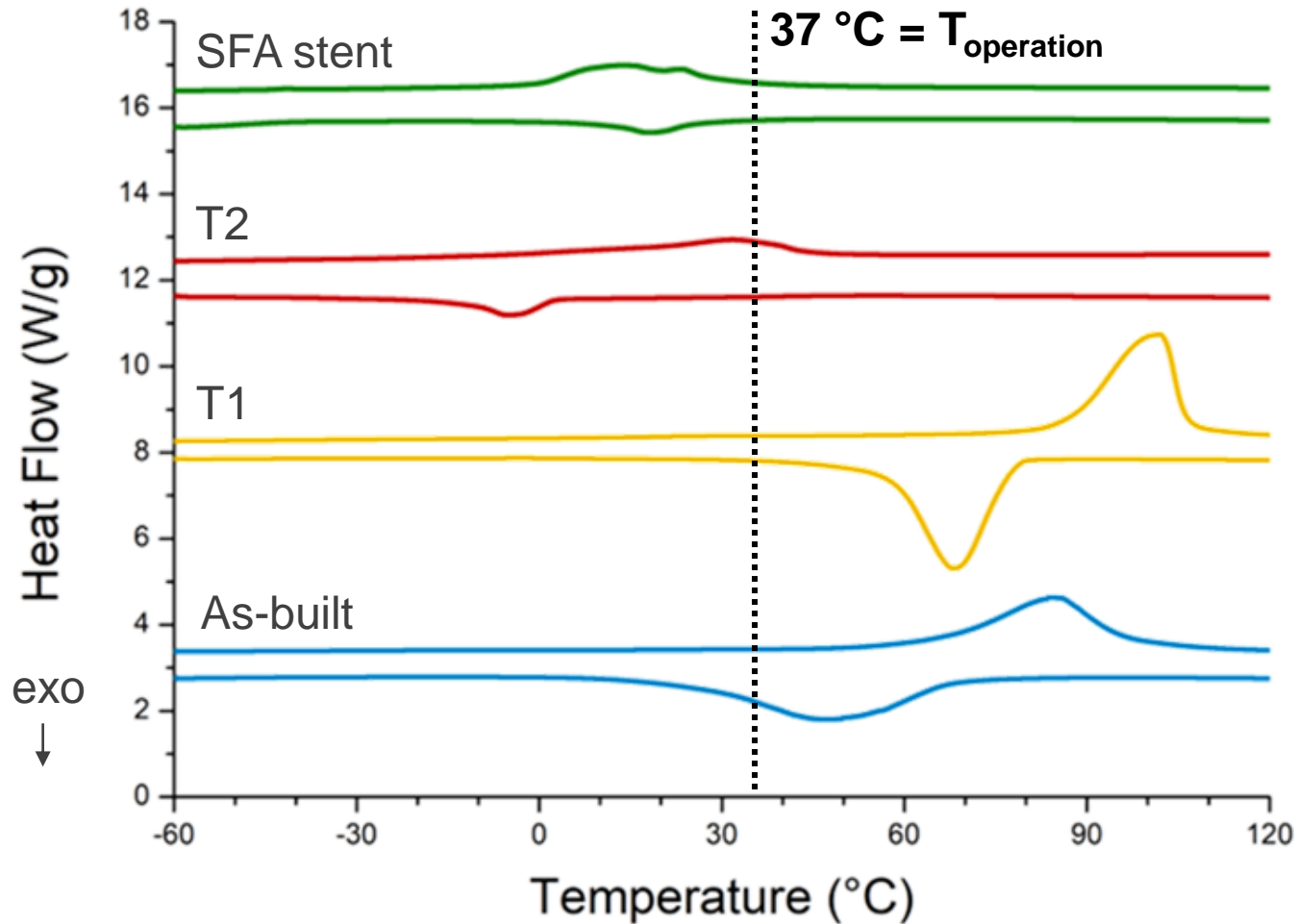
(at%)

RESULTS: HEAT-TREATED SAMPLES



- Permanence of the L-PBF structure
- Martensite phase within the melt pools
- Mixed structure: austenite near the edges, martensite at the center
- Ti_2Ni phase formation
- Nickel rich precipitates ($\approx Ni_3Ti_2$) as result of the matrix chemical variation

RESULTS: HEAT-TREATED SAMPLES



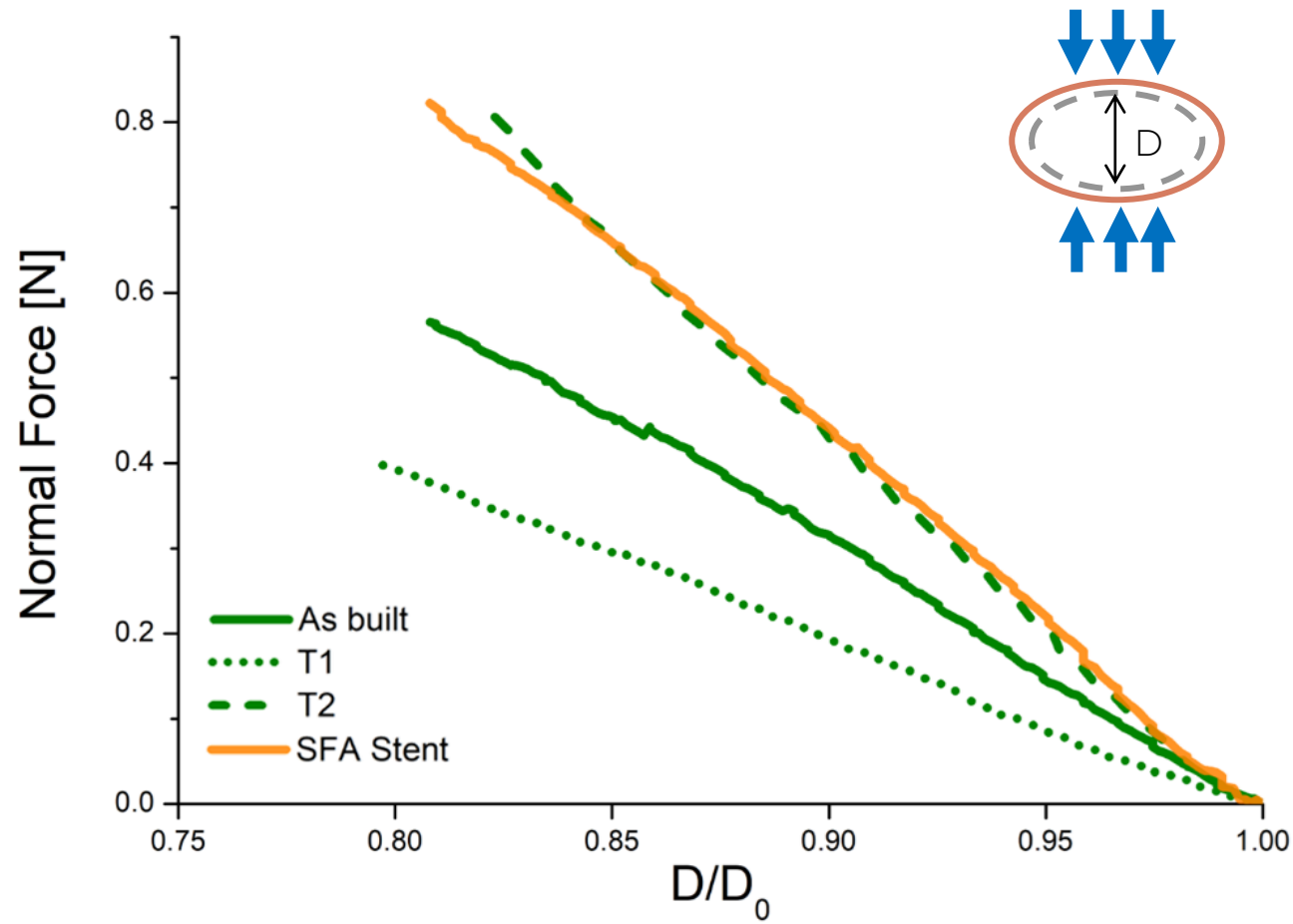
T1

- Slight shift to the right (+20 °C)
- More evident peaks → more homogeneous structure
- Martensite at $T_{\text{operation}}$

T2

- Shift to the left (-60 °C)
- More rounded peaks → more heterogeneous structure
- Mixed structure at $T_{\text{operation}}$
- Curves closer to those of SFA stent

RESULTS: HEAT-TREATED SAMPLES



- T1 worsens resistance
- T2 improves resistance → similar behaviour to SFA stent

Compression resistance (N)	
L-PBF stent	0.70 ± 0.19
T1	0.41 ± 0.01
T2	0.82 ± 0.03
SFA stent	0.82 ± 0.03

CONCLUSIONS

As-built

- 1.5% Nickel loss during the process
- L-PBF stents exhibit the **same overall behaviour**



NEXT DEVELOPMENTS

- Changing the process parameters
- Changing the powder chemical composition

Nitinol powder (%at)	
Ni	Ti
52.39	47.61



Heat-treated

- T1 (600 °C, 1.5 h) → stress relieving
- T2 (1100 °C, 1.5 h) → change in microstructure, which approaches the ideal one for **superelasticity**
→ properties close to those of the conventional stent

Heat treatments can constitute an operative method to shape the alloy to achieve the desired properties, but which in any case cannot disregard from an effective process control

Thanks for the attention

m.abrami003@unibs.it



UNIVERSITÀ
DEGLI STUDI
DI BRESCIA



October 17th–18th, 2022 Plesso Didattico Morgagni, Viale Morgagni, 44-48, 50134 Firenze



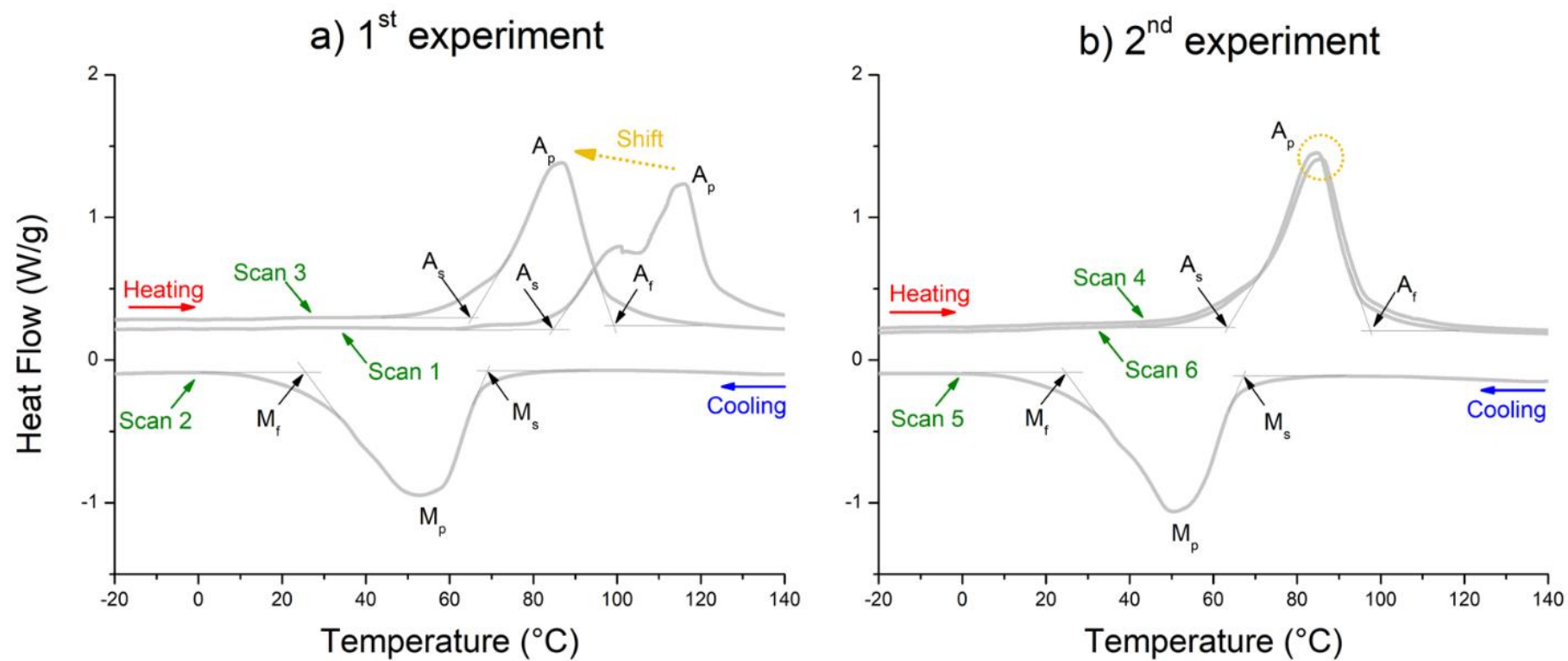
TTs

Material	Temperature (°C)								
	M _f	M _p	M _s	R _f	R _p	R _s	A _s	A _p	A _f
Powder	31	37	49	-	-	-	58	66	77
Sample A	28	53	68	-	-	-	69	87	93
Sample B	35	57	86	-	-	-	71	88	98
Sample C	28	47	68	-	-	-	65	84	97
SFA stent	<-70	-60	-40	10	18	27	19	23	31

Material	Temperature (°C)								
	M _f	M _p	M _s	R _f	R _p	R _s	A _s	A _p	A _f
As built	28	47	68	-	-	-	65	84	97
T1	58	68	78	-	-	-	86	102	106
T2	-13	-5	3	-	-	-	-17	32	45
SFA stent	<-70	-60	-40	10	18	27	-1	14	31



TTs



NEXT DEVELOPMENTS



Nitinol powder (%at)

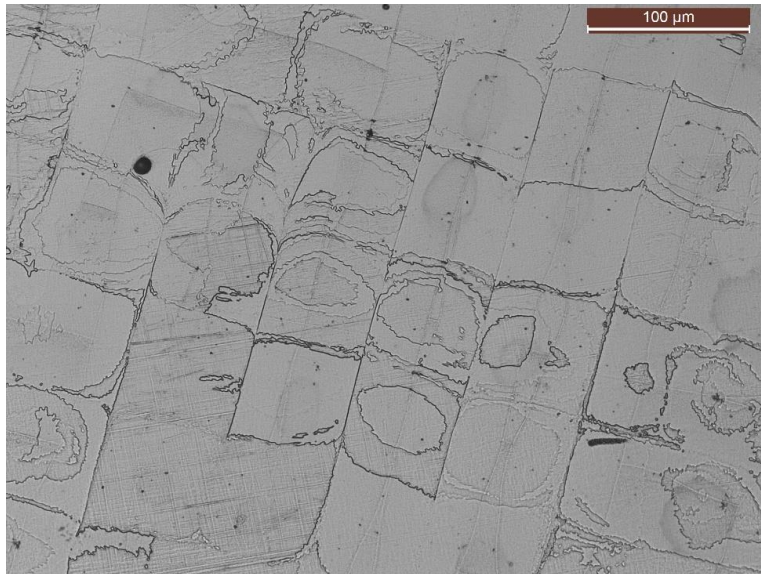
Ni	Ti
52.39	47.61

Sample	Laser Power	Scan Speed	Hatch Space
S2	150 W	500 mm/s	80 μ m
S3	180 W	1000 mm/s	80 μ m
S4	180 W	500 mm/s	80 μ m

Layer thickness = 60 μ m

Spot size = 60 μ m

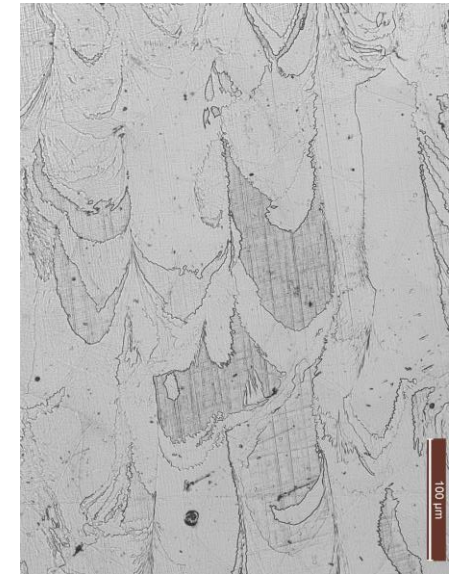
xy, 200X



z, 100X



z, 200X



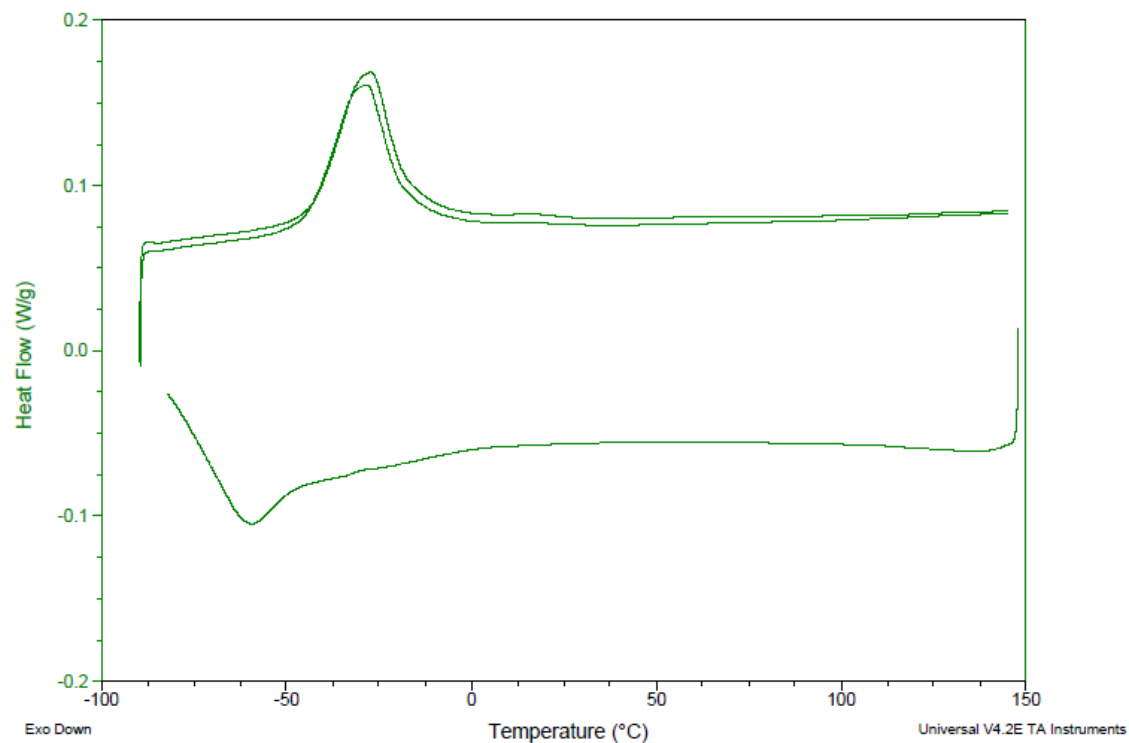
NEXT DEVELOPMENTS



Nitinol powder (%at)	
Ni	Ti
52.39	47.61

Sample	Laser Power	Scan Speed	Hatch Space
S2	150 W	500 mm/s	80 μm
S3	180 W	1000 mm/s	80 μm
S4	180 W	500 mm/s	80 μm

Layer thickness = 60 μm
Spot size = 60 μm



October 17th–18th, 2022 Plesso Didattico Morgagni, Viale Morgagni, 44-48, 50134 Firenze

